We have carried out high-resolution angle-resolved photoemission spectroscopy (ARPES) to study the electronic structure of highly metallic Na$_x$WO$_3$ ($x=0.58$, 0.65, 0.7, and 0.8). The experimentally determined valence-band structure has been compared with the results of an *ab initio* band-structure calculation. While the presence of an impurity band (level) induced by Na doping is often invoked to explain the insulating state found at low concentrations, we find no signature of impurity band (level) in the metallic regime. The states near $E_F$ are populated and the Fermi edge shifts rigidly with increasing electron doping ($x$). The linear dispersion of the conduction band explains the linear variation of thermodynamic properties including the specific heat and magnetic susceptibility. The presence of an electron-like Fermi surface at $I(X)$ and its evolution with increasing Na content and the rigid shift of the Fermi level with increasing $x$ agrees well with the band calculation.

I. INTRODUCTION

The studies of tungsten-oxide-based materials are of enormous interest in material science as a result of various applications. Stoichiometric bulk WO$_3$ is yellowish-green and monoclinic at room temperature. It is possible to insert sodium (Na) in WO$_3$, thus forming the series Na$_x$WO$_3$; the color changes from yellowish-green to gray, blue-deep-violet, red, and finally to gold as $x$ increases from zero to unity.$^1$ WO$_3$ modified by ion incorporation or substoichiometry exhibits many technologically important properties.$^{2,3}$ Na$_x$WO$_3$ exhibits interesting electronic properties, especially a metal-insulator transition (MIT) as a function of $x$. A high-metallic conduction is obtained for $x \approx 0.25$, while the system undergoes MIT with decreasing $x$. Hence, the study of electronic structure of Na$_x$WO$_3$ is of great interest from both technological and fundamental perspectives.

The structural evolution$^4$ of Na$_x$WO$_3$ is also interesting to study, since it changes from monoclinic to orthorhombic, to tetragonal, and finally to cubic with increasing $x$. For $x \approx 0.5$, it exists in a variety of structural modifications, while for $x \approx 0.5$, Na$_x$WO$_3$ is a metallic continuous solid solution with perovskite-type crystal structure. Brown and Banks$^5$ have shown that the lattice parameter varies as $a = 3.7845 + 0.0820x$(Å) for highly metallic Na$_x$WO$_3$. Figure 1(a) shows the crystal structure of NaWO$_3$. In Na$_x$WO$_3$, Na ions occupy the center of the cube, while the W ions are located at the cube corners. The oxygens are at the edge centers and, therefore, form WO$_6$ octahedra with each W ion. The octahedral crystal field of the six oxygen neighbors of the W split the W 5$d$ bands into triply degenerate $t_{2g}$ and doubly degenerate $e_g$ bands (in the cubic phase, when the WO$_6$ octahedra are distorted, the degeneracy of these levels may be further split by the lowering of the symmetry). In WO$_3$, the Fermi level ($E_F$) lies at the top of the O $p$ bands, and WO$_3$ is a band insulator. Within a rigid-band model, the band structure of both WO$_3$ and NaWO$_3$ should be identical with $E_F$ at different positions. In Na$_x$WO$_3$, the Na 3$s$ electrons are transferred into the W 5$d$ $t_{2g}$ band and the system becomes metallic for $x \approx 0.25$. However, for low concentration of $x (x \approx 0.25)$ the material is insulating; the origin of the MIT is still under debate. According to the Anderson localization model,$^7$ the random distribution of Na$^+$ ions in the WO$_3$ lattice gives rise

FIG. 1. (a) Crystal structure of NaWO$_3$ and (b) cubic Brillouin zone of NaWO$_3$ showing high-symmetric lines.
to strong disorder effects that lead to the localization of states in the conduction band tail. The system undergoes a MIT for low Na concentration. An alternative explanation for the MIT is the development of an impurity band (level) induced by Na doping, where the states become localized at low Na concentration. Another possibility driving the MIT is the splitting of the band at the chemical potential (or $E_F$) into two bands: the upper Hubbard band (UHB) and the lower Hubbard band (LHB), where the localization occurs in a pseudogap between the two bands. However, the Hubbard gap would coincide with the chemical potential (or $E_F$) only if the band were half filled, which is unlikely in the context of Na$_x$WO$_3$. Among the various models proposed to explain the MIT in Na$_x$WO$_3$, Anderson localization seems to be the most appropriate one.

Many photoemission studies have been reported on Na$_x$WO$_3$ and Höchst et al. have performed angle-resolved photoemission spectroscopy (ARPES) on a metallic Na$_{0.65}$WO$_3$ single crystal with poor energy and momentum resolution. There are conflicting reports on the mechanisms of MIT as mentioned above in many of the previous studies. The evolution of electronic structure with Na doping, where the states become localized at low Na concentration. An alternative explanation for the MIT in Na$_x$WO$_3$ is the development of an impurity band, which is one of the mechanisms proposed for the MIT in Na$_x$WO$_3$.

In this paper, we report high-resolution ARPES on highly metallic Na$_x$WO$_3$ ($x=0.58$, $0.65$, $0.7$, and $0.8$). The valence-band structure, as well as the Fermi surface (FS) have been determined experimentally. We have also carried out ab initio band-structure calculations based on the plane-wave pseudopotential method and compared the calculated results with experiment. We did not find any impurity band (level) in our band mapping, which is one of the possible mechanisms to explain the observed MIT in Na$_x$WO$_3$. The Fermi surface shows an electron-like pocket centered at the $\Gamma(X)$ point in the Brillouin zone [Fig. 1(b)], in good agreement with the band calculation. The volume of Fermi surface monotonically increases with $x$ from 0.58 to 0.8, indicating that Na $3s$ electrons go to the W $5d$ $t_{2g}$ conduction band and the states near $E_F$ are filled with Na doping. We find that a simple, rigid band shift of the features can explain the $x$-dependent band structure in highly metallic Na$_x$WO$_3$, where the doped electrons merely fill up the conduction band.

II. EXPERIMENTS

Cubic single crystals of Na$_x$WO$_3$ ($x=0.58$, $0.65$, $0.7$, and $0.8$) were grown by the fused salt electrolysis of Na$_2$WO$_4$ and WO$_3$ as described by Shanks. The resistivity measurements and Laue diffraction pattern show that the crystals are metallic with a single cubic phase, respectively. The $x$ values were obtained from the measured lattice parameters as determined experimentally. We have also carried out photoemission spectroscopy measurements with a GAMMADATA-SCIENTA SES 200 spectrometer with a high-flux discharge lamp and a toroidal-grating monochromator. The He I$\alpha$ ($h\nu=21.218$ eV) resonance line was used to excite photoelectrons. The energy and angular (momentum) resolutions were set at $5–11$ meV and $0.2^\circ (0.01$ Å$^{-1}$), respectively. The measurements were performed at 14 K in a vacuum of $3\times10^{-11}$ Torr base pressure. A clean surface for photoemission measurements was obtained by in situ cleaving. After each set of measurements, we checked the degradation of the sample surface by checking the background of the spectrum and found no degradation to the surface. The Fermi level ($E_F$) of the sample was referred to that of a gold film evaporated on the sample substrate.

III. BAND CALCULATIONS

We have performed ab initio band-structure calculations using projected-augmented wave potential as implemented in the VASP code. A k-points mesh of $8\times8\times8$, lattice constant of 3.85 Å, and the generalized-gradient approximation for the exchange was used for the calculation. We have simulated electron doping in our calculations by a rigid band shift of the band structure. The Fermi surfaces have been calculated for different concentrations.

IV. RESULTS AND DISCUSSION

A. Valence-band region

Figure 2 shows the valence-band ARPES spectra of Na$_{0.58}$WO$_3$ and Na$_{0.65}$WO$_3$ measured at 14 K with He I$\alpha$ photons (21.218 eV) along the $\Gamma(X)$-$M(R)$ direction of Brillouin zone.
The intensity near $E_F$ at $\Gamma(X)$ is gradually reduced while moving toward $M(R)$. There are three prominent peaks visible around the $\Gamma(X)$-$M(R)$ direction. A nondispersive peak at 3.3 eV at $\Gamma(X)$ gradually loses its intensity while moving toward $M(R)$ and once again becomes prominent around $M(R)$. The most prominent peak in the valence band is seen at 4.3 eV at $\Gamma(X)$ and disperses downward around $\Gamma(X)$ along with the decrease of its peak intensity. We observe a broad peak at 6.3 eV around $\Gamma(X)$, which disperses downward by going toward $M(R)$. It is to be noted that another weak feature, away from $\Gamma(X)$, is developing around 5 eV with increasing $x$. No additional bands are found to emerge in the relevant energy window with Na doping, which suggests that a rigid-band model is adequate.

Figure 3 shows the valence-band ARPES spectra of Na$_x$WO$_3$ for $x=0.7$ and 0.8, measured at 14 K with He I$\alpha$ photons along both the $\Gamma(X)$-$X(M)$ and $\Gamma(X)$-$M(R)$ high-symmetry lines. We find that all the spectral features are essentially similar in Na$_x$WO$_3$ for both $x=0.7$ and 0.8. However, in $\Gamma(X)$-$X(M)$ direction, the intense peak at 4.3 eV does not disperse as in the $\Gamma(X)$-$M(R)$ direction [see Figs. 3(b) and 3(d)]. The weak spectral feature around 5 eV develops clearly when the Na concentration increases to 0.8. We clearly see four peaks near $\Gamma(X)$ point [Fig. 3(d)]. Two peaks between 4 and 6 eV marked as $A$ and $B$ merge to a single intense peak at 4.3 eV around $\Gamma(X)$. The peak marked $B$ highly disperses downward while moving away from $\Gamma(X)$ and merges with the 6.3 eV peak, whereas the peak marked as $A$ slowly disperses to lower-binding energy. With increasing Na concentration in Na$_x$WO$_3$, all the peaks become more intense and sharper as contrasted in Figs. 2(a) and 3(d). This is attributed to the decrease of disorder in the system.

In Fig. 4, we show the near-valence band-edge region of the valence-band spectra at $\Gamma(X)$ of the different compositions studied here. A clear shift in the valence-band edge is seen as the Na content is increased. We have shifted the spectra (after normalizing the area under the curve), superimposed them, and determined the shift to be $-0.24$ eV as $x$ is varied from 0.58 to 0.8. Figure 4(b) shows the theoretically computed energy shift calculated from a rigid-band model. The experimental shift (0.24 eV) is quite close to the theoretically computed shift of $-0.3$ eV.

We have mapped out the band structure of Na$_x$WO$_3$ for $x=0.58$ and 0.65 along the $\Gamma(X)$-$M(R)$ direction in Figs. 5(a) and 5(b), respectively. The experimental band structure has been obtained by taking the second derivative of the ARPES spectra and plotting the intensity by gradual shading as a function of the wave vector and the binding energy. The bright areas correspond to the experimental bands. We also...
Theoretical band structure of NaWO₃ based on the plane-wave pseudopotential method is also shown by thin, solid and dashed lines for comparison. The experimental band structures are essentially similar to the band calculation, which we have described in a later part of this article. The experimental band structures are essentially similar for both x = 0.58 and 0.65 along the Γ′(X)-M(R) direction. The top of the valence band (band P) at 3.3 eV binding energy arises from the second derivative of the ARPES intensity. But we clearly see the dispersion in near-Γ′(X) band mapping, which we have described in a later part of this article.

The dispersion of band P around 4.3 eV binding energy at Γ′(X) and another band S, which disperses highly downward from Γ′(X) to M(R) are in good agreement with the band calculation. The intensity along the Γ′(X)-M(R) direction at 4.3 eV arises due to finite-electron scattering from the strong intensity of bands P and S at Γ′(X) and vanishes with the decrease of disorder in NaₓWO₃ (for x = 0.7 and 0.8 disorder is less; hence, the effect is not clearly visible). This flat band raises questions on the homogeneity of the samples; hence, we have carried out various measurements to ensure the homogeneity of the samples for all the compositions studied here. The x-ray diffraction (XRD) pattern of the single crystals shows that the samples are of good quality without impurity. The energy dispersive x-ray analysis (EDAX) measurements also do not show any inhomogeneity within the instrument resolution. Now, repeated charge-coupled device (CCD) XRD (however, CCD XRD is nothing but single-crystal diffraction experiments in which the data are collected simultaneously from many angles and, therefore, the collection time is short) measurements on our samples showed that the crystals are of good quality (with sharp spot intensities), without any defect (like twin boundaries, etc.), and devoid of any impurity (no extra spots are found and the pattern in all could be matched with the standard cubic structure of metallic NaₓWO₃). Hence, we conclude that the flat band is arising from angle-integrated background due to the presence of finite disorder in the samples. From the band calculation, we assign the broad band at 3.5 and 4.5 eV around X₃–₉ eV to a group of highly dispersive bands. The gross features of the experimental valence band at higher-binding energy (4.8–5 eV) can be explained by the ab initio band-structure calculations. These calculations also show that the valence band (3.9 eV) consists of mostly O 2p character of NaₓWO₃ with a small admixture of bonding W 5d character.

Figure 6 shows the experimental band mapping of NaₓWO₃ for x = 0.7 and 0.8 along the Γ′(X)-X(M) and Γ′(X)-M(R) high-symmetry lines, together with the ab initio band structure. Both band structures for x = 0.7 and 0.8 along Γ′(X)-X(M) and Γ′(X)-M(R) are essentially similar. In both compositions, we find two, flat nondispersive bands (marked Π and Ω) at 3.5 and 4.5 eV around Γ′(X) along the Γ′(X)-X(M) direction. The energy position of the experimental bands does not agree well with the band calculation, al-

FIG. 5. Experimental valence-band structure of (a) Na₀.₅₈WO₃ and (b) Na₀.₆₅WO₃ obtained from the ARPES experiment along the Γ′(X)-M(R) direction. Bright areas correspond to the experimental bands and dark, dashed lines are guides to the experimental bands. The experimental band structure of NaWO₃ based on the plane-wave pseudopotential method is also shown by thin, solid and dashed lines for comparison.

FIG. 6. Experimental valence-band structure of Na₀.₃₅WO₃ (a) along Γ′(X)-X(M) and (b) along Γ′(X)-M(R) and of Na₀.₄₅WO₃ (c) along Γ′(X)-X(M) and (d) along Γ′(X)-M(R) direction. Bright areas correspond to the experimental bands and dark, dashed lines are guide to the experimental bands. Theoretical band structure of NaWO₃ based on the plane-wave pseudopotential method is also shown by thin, solid and dashed lines in respective high-symmetry directions for comparison.
though the band calculation predicts a flat band along the \( \Gamma(X)-X(M) \) direction. The band structures along the \( \Gamma(X)-M(R) \) direction for \( x=0.7 \) and 0.8 are similar to those of \( x=0.58 \) and 0.65, showing four dispersive bands in the valence-band region. Comparison of valence-band structures of \( x=0.58 \) and 0.8 shows that all the bands in the valence-band regime move downward rigidly. Thus, with increasing \( x \) in Na\(_{x}\)WO\(_3\), the Na 3\( s \) electrons just fill the W 5\( d \) \( t_{2g} \) conduction band and change the \( E_F \) position consistent with the rigid-band model appropriate for the compositions studied here.

B. Near-\( E_F \) region

In order to study the electronic structure near \( E_F \) in more detail, we have carried out high-resolution ARPES measurements with a smaller energy interval and a higher signal-to-noise ratio. Figures 7 and 8 show the high-resolution ARPES spectra near \( E_F \) of Na\(_{x}\)WO\(_3\) for \( x=0.58, 0.65, 0.7, \) and 0.8 measured at 14 K with He I\( \alpha \) photons along the high-symmetry directions in the Brillouin zone. We observe a very weak broad feature near 0.9 eV at \( \Gamma(X) \), which disperse upward to form an electron-like pocket at \( \Gamma(X) \) for all compositions of \( x \). There is no signature of such a feature at \( X(M) \) or \( M(R) \). As the Na concentration increases, this feature becomes very prominent as shown in Fig. 8(d). This behavior may be due to the decrease of disorder with increasing \( x \) in the system. In both figures, the bottom of the conduction band lies roughly around 0.9–1.0 eV below \( E_F \). The exact position of the band bottom is difficult to determine due to the very low spectral intensity at the band bottom; nevertheless, it is clear that the conduction band bottom moves downward with Na concentration similar to the trend seen for the valence band. This can be explained by considering the simple rigid-band shift. Since the determination of the exact position of the band bottom has much more ambiguity, it is difficult to determine quantitatively the shift of the band bottom from \( x=0.58 \) to 0.8. Previous reports\(^{13,21–23}\) that the bandwidths of occupied conduction states appear to be al-

FIG. 7. ARPES spectra near \( E_F \) of (a) Na\(_{0.58}\)WO\(_3\) and (b) Na\(_{0.65}\)WO\(_3\) measured at 14 K with He-I\( \alpha \) photons (21.218 eV) along the \( \Gamma(X)-M(R) \) direction. Vertical bars are a guide to the eye for band dispersion.

most independent of Na concentration are not supported by our results. Since the density of states (DOS) at band bottom is very low and the background is high, it is impossible to get the exact bandwidths of occupied states from the previous angle-integrated measurements.

Figure 9 shows the plot of ARPES intensity as a function of the wave vector and the binding energy, showing the experimental band structure near \( E_F \). We find an electron-like pocket at \( \Gamma(X) \), whose linear dispersion at \( E_F \) agrees satisfactorily with the band calculation. We find a clear variation in the spectral intensity at \( E_F \), which suggests that the band crosses \( E_F \) at the highest intensity. Though our present study on different compositions are far from MIT regime, we believe that if the impurity band and/or level would be the cause of MIT, then it should have signature in metallic regime though the impurity band and/or level would have lain much below the Fermi level. But no such signature of the impurity band (level) near \( E_F \) is seen in Fig. 9 except W 5\( d \) bands, which rules out the development of a Na-induced impurity band (level). Hence, we conclude that the previous speculation regarding MIT at low Na concentration being due to the development of Na-induced impurity band (level) is not supported by our results.

FIG. 8. ARPES spectra near \( E_F \) of Na\(_{0.7}\)WO\(_3\) (a) along \( \Gamma(X)-X(M) \) and (b) along \( \Gamma(X)-M(R) \) and of Na\(_{0.8}\)WO\(_3\) (c) along \( \Gamma(X)-X(M) \) and (d) along \( \Gamma(X)-M(R) \) direction, measured at 14 K with He-I\( \alpha \) photons (21.218 eV). Vertical bars are a guide to the eye for band dispersion.
In the rigid-band model with a spherical Fermi surface and rigid parabolic density of states, the density of states \( N(E) \) is proportional to \( E^{1/2} \). It is assumed that all sodium atoms are ionized in \( \text{Na}_x\text{WO}_3 \); hence, \( N(E_F) \), the density of states at \( E_F \), is proportional to \( x^{1/3} \). However, the physical properties including the magnetic susceptibility and the specific heat coefficient \( \gamma \) were found to vary linearly with \( x \). We extrapolated the band dispersion from the highest intensity points of the band mapping shown as open circles in Fig. 9 and find that the conduction bandwidth expands with increasing \( x \); the experimental band dispersion is not a free electron-like parabolic as proposed before. The \textit{ab initio} band structure results also show linear band dispersion as observed experimentally. The expansion of the conduction band can be well explained by the linear increase in the density of states of the conduction band with increasing \( x \) in \( \text{Na}_x\text{WO}_3 \). This explanation fits well with the \( x \)-dependent behavior of the specific heat and the magnetic susceptibility, which vary linearly with \( x \) for cubic, metallic \( \text{Na}_x\text{WO}_3 \). The heat capacity data \(^{24} \) show that the effective mass \((m^*)\) of conduction electrons increase monotonically with \( x \). We determined the effective mass from the Fermi velocity \( v_F \) at \( E_F \) and found a similar monotonic increase in the effective mass of the conduction electrons for all \( x \) values studied here. The band mass is less than the free-electron mass \( m_0 \) and agrees well quantitatively with the mass found from other experiments.\(^{25} \)

### C. Fermi surface topology

In Fig. 10, we show the ARPES-intensity plot at \( E_F \) for \( \text{Na}_x\text{WO}_3 \) (\( x=0.58, 0.7, \) and 0.8) as a function of the two-dimensional (2D) wave vector. The intensity is obtained by integrating the spectral weight within 20 meV with respect to \( E_F \) and symmetrized with the cubic symmetry. We have also calculated the FS(s) (on \( \Gamma X M X \) and \( X M R M \) plane) for fractional Na concentration in \( \text{Na}_x\text{WO}_3 \) assuming rigid band shifts, which are shown by dotted lines. We observe one spherical electron-like Fermi surface centered at the \( \Gamma \) point, which is covered with another square-like Fermi surface. Along the \( \Gamma(X)-X(M) \) direction, we find only one \( k_F \) point; while along the \( \Gamma(X)-M(R) \) direction, there are two distinct \( k_F \) points. These two Fermi surfaces are attributed to the W 5d \( t_{2g} \) bands. On increasing the Na concentration, the Na 3s electrons are transferred to the W 5d \( t_{2g} \) band at \( E_F \). Hence, the volume of the FS gradually increases in accordance with the increase of the Na concentration. The 2D area of the FS at \( \Gamma(X) \) is estimated to be \( 11.5 \pm 2\% \) of the whole.
\(H\)

\(XMX\) Brillouin-zone 2D plane in \(Na_{0.58}WO_3\) and increases to 16.8±2% in \(Na_{0.8}WO_3\) [see Fig. 10(c)]. Whereas, we found the calculated 2D area of FS of \(NaWO_3\) at \(\Gamma(X)\) to be 22% of the whole \(\Gamma XMX\) Brillouin-zone plane. The square-like FS centered at the \(M\) point is not clearly visible in \(Na_{x}WO_3\) for \(x=0.58\) and 0.7, but is prominent in \(x=0.8\). This FS arises from one single band, whereas the FSs observed at \(\Gamma(X)\) are from three bands (two from \(\Gamma XMX\) and one from the \(XMRM\) planes); hence, the intensity is much more enhanced around \(\Gamma(X)\) in the ARPES experiment. The volume of calculated FS and experimental FS matches well for all compositions studied here. A rigid shift of the Fermi energy is found to give a qualitatively good description of the Fermi surface.

V. CONCLUSION

We have carried out high-resolution angle-resolved spectroscopy on highly metallic \(Na_xWO_3\) for \(x=0.58, 0.65, 0.7,\) and 0.8. We have experimentally determined the valence-band structure and compared it with the \textit{ab initio} band-structure calculations. From the comparison of the experimental band structure of \(Na_{0.58}WO_3\) and \(Na_{0.8}WO_3\), we concluded that the rigid shift of band structure can explain the metallic \(Na_xWO_3\) band structure with respect to Na doping. We did not observe any signature of impurity band (level) near \(E_F\), and hence, the possibility of development of Na-induced impurity band (level) is ruled out. We observed linear band dispersion near \(E_F\), which shows that the conduction band is not a free electron-like parabolic band. The band mass increases linearly and well explains the linear behavior of thermodynamic properties with increasing Na doping. We observed electron-like FS at the \(\Gamma(X)\) point as predicted from band calculation and the FS gradually increases with increasing Na concentration due to W 5\(d_{\ell 2}\) band filling. A rigid shift of the Fermi energy is found to give a qualitatively good description of the Fermi surface.

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